# Serine-Based Homochiral Nanoporous Frameworks for Selective CO<sub>2</sub> Uptake

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The reduction of  $CO_2$  in the air is now a worldwide task to

avoid the greenhouse effect.<sup>1</sup> Selective trapping of CO<sub>2</sub> from air

or the emissions of coal-fired power plants under ambient

conditions becomes more and more important for this purpose.<sup>2</sup>

Recently, porous metal organic framework materials (MOFs)

have been considered as very promising candidate for gas storage

and separation because of their high surface area and tunable

structural features.<sup>3-5</sup> The storage of CO<sub>2</sub> and its selective separation from gas mixtures are two main tasks for MOF

materials. Of particular interest is the efficient separation of

CO<sub>2</sub> from N<sub>2</sub> at ambient temperature and pressure, because that

is the key fact in the capture of carbon from flue gas. Numbers of

MOF materials with selective adsorption of CO<sub>2</sub> over N<sub>2</sub> have

been reported in near five years, but highly efficient materials are

still rare emerging.<sup>6,7</sup> So far how to enhance the separation ability of the materials from structural modification approach

remains great challenge. The current research efforts in this area have focused on either (1) increasing pore volume or

surface area to improve  $CO_2$  capacity or (2) modifying the

physical and chemical environments in pores by introducing functional moieties with high affinity to  $CO_2$ .<sup>8,9</sup> It has been

recognized that some specific polar groups, for example,

-OH, -COOH, or -NH<sub>2</sub>, can dramatically help on the selec-

tivity of CO<sub>2</sub> rather than other gases (e.g.,  $N_2$ , CH<sub>4</sub>, H<sub>2</sub>).<sup>10</sup>

Considering the above points, what should happen if a

combination of such specific polar groups in a molecular

building block was realized by the porous MOF material? Unfortunately, the answer is still unknown and no related

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Supporting Information

INTRODUCTION

**ABSTRACT:** Presented here are two serine-based homochiral materials that show isostructural nanoporous three-dimensional frameworks with high selectivity and storage capacity for  $CO_2$  over  $N_2$  and CO at ambient conditions.

# A special molecule containing all -OH, -COOH, and $-NH_2$ groups is serine, one of the typical amino acid. However, porous three-dimensional MOF materials based on serine have never been explored to date, although other amino acid-based porous MOFs have been reported.<sup>11</sup> Serine is such a functional ligand not only due to its multidentate coordinate sites, but also due to its intrinsic homochirality. MOFs constructed from commercial available enantiopure serine should take another outstanding merit on homochiral applications, such as enantioselective catalysis and separation.<sup>11–13</sup> Since the presence of rich polar groups, serine-based porous MOF materials may open a new approach toward the CO<sub>2</sub> capture and its selective separation.

Herein, we report two homochiral materials synthesized from D- or L-serine, respectively, which show isostructural nanoporous three-dimensional frameworks with high selectivity and storage capacity for carbon dioxide over nitrogen at ambient conditions.

### EXPERIMENTAL SECTION

**General Procedures.** All the syntheses were performed in 20 mL glass vial under autogenous pressure. Reagents were purchased commercially and used without further purification. Thermal analysis was carried out on a Netzsch STA449C thermal analyzer under dinitrogen atmosphere with a heating rate of 10  $^{\circ}$ C·min<sup>-1</sup>. X-ray powder diffraction experiments were performed in a Rigaku Dmax 2500 instrument with an ultra 18Kw Cu rotating anode point source. Gas adsorption measurement was performed in the ASAP (Accelerated Surface Area and Porosimetry) 2020 System.

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porous MOF materials are synthesized.



CO2/N2

CO2/CO

N<sub>2</sub> or CO

Table 1.	Crystal Data	and Structure	Refinement f	for 1-L and
1-D				

compound reference	1-L	1-D	
chemical formula	$c_{22}h_{14}n_2o_{14}zn_4\\$	$c_{22}h_{14}n_2o_{14}zn_4\\$	
formula mass	791.83	791.83	
crystal system	monoclinic	monoclinic	
space group	P2(1)	P2(1)	
a (Å)	19.293(4)	19.420(13)	
b (Å)	14.332(3)	14.437(10)	
c (Å)	10.3246(14)	10.337(7)	
$\alpha$ (deg)	90.00	90.00	
$\beta$ (deg)	90.114(10)	89.731(13)	
$\gamma$ (deg)	90.00	90.00	
unit cell volume	2854.8(9)	2898(3)	
temperature (K)	293(2)	293(2)	
Ζ	2	2	
no. of reflections measured	23 529	24 911	
no. of independent reflections	10 661	12 843	
R <sub>int</sub>	0.0749	0.0639	
final <i>R</i> 1 values $(I > 2\sigma(I))$	0.0494	0.0713	
final wR( $F^2$ ) values ( $I > 2\sigma(I)$ )	0.1235	0.1904	
goodness of fit on $F^2$	0.864	0.955	
Flack parameter	-0.08(5)	0.00(2)	

Synthesis of  $Zn_4(\iota-Ser)_2(bdc) \cdot 6DMF$  (**1-L**). A mixture of  $Zn(NO_3)_2 \cdot 6H_2O$  (0.595 g, 2 mmol),  $\iota$ -serine ( $\iota$ -Ser, 0.105 g, 1 mmol), terephthalic acid ( $H_2bdc$ , 0.166 g, 1 mmol), and 10 mL of *N*,*N*-dimethylformamide (DMF) was heated up and stirred to give a yellow clear solution, and then transferred and sealed in a 23 mL of Teflon-lined airtight reactor was heated at 120 °C for 2 days, and then cooled to room-temperature. Fresh DMF was used to wash the primrose yellow crystals, and the obtained crystals were stored in a desiccator.

Synthesis of  $Zn_4(D-Ser)_2(bdc) \cdot 6DMF$  (**1-D**). **1-D** was prepared by a procedure similar with that of **1-L**, except for the replacement of L-serine by D-serine.

X-ray Crystallography. The diffraction data was collected on a Saturn 70 charge-coupled device diffractometer equipped with confocalmonochromated Mo K $\alpha$  radiation ( $\lambda = 0.71073$  Å) at room temperature. The CrystalClear program was used for absorption correction. The structures were solved by direct methods and refined on  $F^2$  by full-matrix, least-squares methods using the SHELXL-97 program package. All nonhydrogen atoms were refined with anisotropic displacement parameters. The total interstitial solvent molecule contents of 1 were determined by TGA. Crystal data for 1-L and 1-D are listed in Table 1.

## RESULTS AND DISCUSSION

Compounds were synthesized, respectively, by mixing serine (D- or L-form), 1,4-benzenebicarboxlic acid (H<sub>2</sub>bdc), and Zn- $(NO_3)_2 \cdot 6H_2O$  in a 1:1:2 molar ratio in DMF solution at 120 °C for 2 days. Single-crystal X-ray Diffraction studies reveal that they are isostructural and formulated as  $[Zn_4(bdc)_2(L-ser)_2)] \cdot 6DMF$  (1-L) and  $[Zn_4(bdc)_2(D-Ser)_2)] \cdot 6DMF$  (1-D), respectively. The phase purity of the bulk product was independently confirmed by powder X-ray diffraction (XRD).

**Crystal Sructure Description.** The prominent structural feature of 1-L is that the homochiral chains built from enantiopure L-serine and Zn centers are linked together by achiral bdc ligands to form a rod-and-spacer motif with large rhombic channels. Compound 1-L crystallized in the chiral P2<sub>1</sub> space



**Figure 1.** (a) Coordination environment of Zn atom in 1-L. (b) The 3D porous framework viewed along the c axis.



Figure 2.  $N_2$  and  $CO_2$  adsorption isotherms: (a)  $N_2$  at 77 K for 1-L-ht, (b)  $N_2$  at 77 K for 1'-L-ht, (c)  $CO_2$  at 195 K for 1-L-ht, and (d) the hysteresis loop of  $CO_2$  at 195 K for 1-L-ht.

group because of the derivation from the enantiopure L-serine. It is notable that the coordination fashion of L-serine in the structure is unprecedented, because it is a dianion owing to the deprotonation of the hydroxy group and acts as an unusual  $\mu_4$ ligand to bound four  $Zn^{2+}$  ions (Figure 1a). The interesting point is that the  $-NH_2$  group of L-serine bridges two Zn centers with the Zn···Zn distance being 3.52 Å. Each five-coordinate Zn center (ZnO<sub>4</sub>N) has distorted trigonal bipyramid geometry, and they are connected by the L-serine ligands into a homochiral chain along the *c* axis (Figure 1a). Meanwhile, the achiral bdc

compound	$BET^{a}$	CO <sub>2</sub> @273 K <sup>b</sup>	CO <sub>2</sub> @298 K <sup>b</sup>	$Q_{st}(CO_2)^c$	$S(CO_2/N_2)@273 K^d$	H <sub>2</sub> @77 K <sup>b</sup>	H <sub>2</sub> @87 K <sup>b</sup>	$Q_{st}(H_2)^c$	
1-L-ht	136.9	55.5	37.1	32.1	185	105.8	79.8	8.2	
1'-L-ht	224.7	43.7	30.3	28.6	17.5	66.7	50.7	7.5	
<sup><i>a</i></sup> Surface area, m <sup>2</sup> /g. <sup><i>b</i></sup> Adsorptive capability, mL/g. <sup><i>c</i></sup> Isosteric heat of adsorption at 1 bar, kJ/mol. <sup><i>d</i></sup> Selectivity of CO <sub>2</sub> over N <sub>2</sub> , v/v.									

Table 2. Adsorption Data and Isosteric Heat Data for 1-L-ht and 1-L-ht

ligands, whose oxygen atoms occupy the equatorial positions of the homochiral chains, further bridge these parallel homochiral chains to form a neutral 3D framework with large channels along the *c* axis (Figure 1b). The effective size of this rhombic window for each channel is about 20  $\times$  10 Å<sup>2</sup> by measuring the diagonal centers. A large void is generated within the channels with the V<sub>void</sub> without guest DMF molecules of 63.3% as calculated by PLATON.<sup>14</sup> The channels are occupied by the structural disordered DMF molecules. There are six DMF guest molecules per formula unit of the host, as evidenced by thermal gravimetric (TG) analysis data. Compound 1-D prepared from D-serine is the enantiomorphism of 1-L and has similar homochiral structure (space group:  $P2_1$ ) to 1-L. The framework structure of 1-L is also similar to the reported compound  $[Zn_2(bdc)(L-lactate)-$ (DMF)] • DMF where the L-lactate ligand was used and each Zn center is coordinated by one solvent DMF molecule.<sup>12</sup> In 1-L, the -OH group of L-serine acted as a tailor and replaced the coordination site of DMF in that reported compound. Thus it becomes much easier for 1-L to remove the solvent DMF molecules from the pores.

Thermal Stability. Thermal gravimetric analyses (TGA) and X-ray powder diffraction pattern (XRPD) measurements were carried out to examine the thermal stability of this nanoporous framework. The TGA curve of 1-L shows that the trapped solvent DMF molecules can be drastically removed at 250 °C with a weight loss of 35.4% (calculated 35.6%) (Supporting Information Figure S2a). The hollow phase (1-L-ht) of 1-L was obtained after heated at 180 °C for 7 h under high vacuum ( $10^{-8}$  mbar), the powder X-ray diffraction pattern of 1-L-ht have appeared obvious change (Supporting Information Figure S3c). It may be due to the laxity of 1-L after the removal of DMF guests. To attain a more complete evacuation and protect the structural integrality, DMF was exchanged with methanol by soaking crystals of 1-L in methanol for one week. As indicated by TGA and infrared spectroscopy, the methanol activated material 1'-L is DMF free (Supporting Information Figures S2b and S4). Its TGA curve showed that methanol was released below 100 °C. Its hollow phase (1'-L-ht) was obtained after heated at 80 °C for 7 h under high vacuum.

**Gas Sorption Properties.** The permanent porosity of 1-L-ht was established by reversible gas sorption experiments using  $N_2$  and  $CO_2$  at 77 and 195 K, respectively, which all show a typical type I behavior characterized by a plateau reached at low relative pressure indicating the presence of permanent micropores in 1-L-ht (Figure 2 and Table 2). The lower uptake of  $N_2$  (42 cm<sup>3</sup>/g) and higher uptake of  $CO_2$  (127 cm<sup>3</sup>/g) may be induced by the strong polar environment in the pores.  $N_2$  adsorption data gives a BET surface area of 136.9 m<sup>2</sup>/g and a Langmuir surface area of 180.4 m<sup>2</sup>/g. A single data point at relative pressure 0.295 gives a maxima pore volume of 0.063 cm<sup>3</sup>/g by the Horvath–Kawazoe equation. Comparatively, the maximum  $N_2$  uptakes at 1 bar for 1'-L-ht are 92.3 cm<sup>3</sup>/g (STP), giving a BET surface area of 224.7 m<sup>2</sup>/g and a Langmuir surface area of all much larger than that for 1-L-ht. A single data point at relative



Figure 3.  $N_2$  and  $CO_2$  adsorption isotherms: (a)  $CO_2$  at 273 K for 1-Lht, (b)  $CO_2$  at 298 K for 1-L-ht, (c)  $CO_2$  at 273 K for 1'-L-ht, (d)  $CO_2$  at 298 K for 1'-L-ht, (e)  $N_2$  at 273 K for 1-L-ht and (f)  $N_2$  at 273 K for 1'-L-ht.

pressure 0.01 gives a maxima pore volume of  $0.081 \text{ cm}^3/\text{g}$  by the Horvath–Kawazoe equation. The results demonstrate the importance of activation for porous materials.

It is worth noting that there is a small hysteresis loop for the CO<sub>2</sub> sorption of 1-L-ht, suggesting relatively strong adsorbentadsorbate interactions between the framework and  $CO_2$ .<sup>15</sup> This phenomenon encourages us to further explore its potential properties on  $CO_2/N_2$  gas separation under ambient conditions. The adsorption isotherms of CO<sub>2</sub> and N<sub>2</sub> for 1-L-ht and 1'-L-ht at 273 K were measured, respectively (Figure 3, Table 2). The uptake value of  $CO_2$  for 1-L-ht was 55.5 cm<sup>3</sup>/g (2.49 mmol/g and 109.2 mg/g, which is comparable to the well-known MOF-5 and larger than many other reported compounds.<sup>7a</sup> However, the CO<sub>2</sub> uptake of 43.7 cm<sup>3</sup>/g for 1'-L-ht is much less than that of 1-L-ht. In particular, CO<sub>2</sub> uptake (42.1 mg/g at 273 K) of 1-Lht at 0.1 bar, a typical partial pressure of CO<sub>2</sub> in flue gases from power plants and steel mills, is comparable or even superior to that of many MOFs.<sup>2</sup> To better understand these observations, the adsorptive capability of 1-L-ht and 1'-L-ht were measured under 298 K, giving an uptake of 37.1 cm<sup>3</sup>/g for 1-L-ht and 21.4 cm<sup>3</sup>/g for 1'-L-ht (Figure 3b and 3d). The enthalpy of  $CO_2$ adsorption for 1-L-ht was estimated from the sorption isotherms at 273 and 298 K using the virial equation to understand the strong affinity of 1-L-ht toward CO<sub>2</sub> (Supporting Information Figures S5 and S6). At zero coverage, the enthalpy of CO<sub>2</sub> adsorption is 32.1 kJ/mol for 1-L-ht and 28.6 kJ/mol for 1'-L-ht, which is comparable to those of MOFs with organic ammonium ions in the pores for strong CO<sub>2</sub> binding.<sup>9b</sup> In contrast, N<sub>2</sub> was hardly adsorbed at all under the conditions of 273 K (just 0.3 cm<sup>3</sup>/g for 1-L-ht and 2.5 cm<sup>3</sup>/g for 1'-L-ht). The maximal uptakes of the CO<sub>2</sub> and N<sub>2</sub> at 273 K and 1 bar for 1-L-ht were used to estimate the adsorption selectivity for CO<sub>2</sub> over  $N_2$ .<sup>16</sup> From these data, the calculated  $CO_2/N_2$  selectivity is 185:1 for 1-L-ht and 17.5:1 for 1'-L-ht at 273 K. MOFs with such high CO<sub>2</sub>/N<sub>2</sub> adsorption selectivity at ambient conditions were rarely reported.<sup>17</sup>



Figure 4. High-pressure adsorption isotherms of  $CO_2$  and CO at 298 K for 1-L-ht.



Figure 5.  $H_2$  adsorption isotherms: (a)  $H_2$  at 77 K for 1-L-ht, (b)  $H_2$  at 87 K for 1-L-ht, (c)  $H_2$  at 77 K for 1'-L-ht, (d)  $H_2$  at 87 K for 1'-L-ht.

In view of the structural feature of 1-L, such gas-adsorption selectivity might be associated with the presence the  $\pi$ -electrons of the ligands and the polar  $-NH_2$  groups, giving rise to an electric field to induce a dipole in CO<sub>2</sub> and an interaction between CO<sub>2</sub> and  $-NH_2$ .<sup>10</sup> The presence of the  $\mu_2$ -NH<sub>2</sub> groups and C=O groups of neighboring ligands provides chances on the inteaction between of CO<sub>2</sub> and  $-NH_2$ .<sup>9c</sup> These cooperative effects yield an enhanced lone pair polarization of the CO<sub>2</sub> electron density, highlighting the impact of the specific pore geometry in the structure of 1-L.

The high-press sorption experiments of  $CO_2$  and CO were also measured at 298 K for 1-L-ht, which show distinct sorption behavior. The  $CO_2$  uptake by 1-L-ht at 40 bar can reach 84 cm<sup>3</sup>/ g (3.75 mmol/g) and are not fully saturated, while the saturated uptake of CO is limited to 22 cm<sup>3</sup>/g (0.98 mmol/g) at 26 bar (Figure 4). Interestingly, there is significant hysteresis for  $CO_2$ sorption whereas it is opposite for CO sorption. The significant hysteresis for 1-L-ht may be attributed to the dynamic feature of the host framework. The selective adsorption of  $CO_2$  over CO may be attributed to the sorbate—sorbent affinity between the  $CO_2$  molecules and the polar framework.

Low-pressure  $H_2$  adsorption isotherms collected for samples of 1-L-ht and 1'-L-ht indicated that the framework had a strong affinity for binding  $H_2$  (Table 2 and Figure 5). At 77 K and 1.06 bar (800 Torr), a fully reversible uptake of 105.8 mL/g  $H_2$  for 1-L-ht but 66.7 cm<sup>3</sup>/g  $H_2$  for 1'-L-ht are apparent. As a further test, a second  $H_2$  adsorption isotherm was measured at 87 K, and the two data sets were used to determine the isosteric heat of  $H_2$ adsorption. The enthalpy of  $H_2$  adsorption for 1-L-ht and 1'-L-ht and was estimated from the sorption isotherms at 77 and 87 K



Figure 6.  $H_2$  adsorption isotherm for 1-L-ht recorded at 77 K, and the red solid line is the fitting of the  $H_2$  adsorption isotherm of 1-L-ht using Langmuir—Freundlich equation.



Figure 7. CH<sub>4</sub> adsorption isotherm for 1-L-ht recorded at 273 K.

using the virial equation to understand the affinity of 1-L-ht and 1'-L-ht toward H<sub>2</sub> (Supporting Information Figure S8–10). The isosteric heat of H<sub>2</sub> adsorption for 1-L-ht and 1'-L-ht reach the values of 8.2 and 7.5 kJ/mol.

What is more, the gravimetric  $H_2$  and  $CH_4$  sorption isotherms of **1-L-ht** were also measured at 77 and 273 K, respectively. The  $H_2$  sorption on **1-L-ht** showed type-I sorption behavior indicatives of the filling of the micropores only. The capacity of **1-L-ht** to adsorb  $H_2$  increases to a saturation of 1.87 wt % at 32 bar (Figure 6), and the adsorption amounts of  $CH_4$  is 5.64 wt % at 6 bar (Figure 7). In addition, the density of the maximum adsorbed  $H_2$  (2.77 wt %, obtained by fitting of the Langmuir—Freundlich (L—F) equation to the adsorption isotherm.) is much higher than the experimental value (1.87 wt %).

### CONCLUSION

In summary, two serine-based homochiral nanoporous framework materials were successfully synthesized and 1-L exhibited high selectivity and storage capacity for  $CO_2$  over  $N_2$  and CO at ambient conditions. The results indicated the special coordination mode of serine ligand and its functionality on  $CO_2$  uptake. This work also revealed that a flexible framework material with suitable pore size and polar functional groups may be a good candidate for the application in  $CO_2/N_2$  separation.

### ASSOCIATED CONTENT

**Supporting Information.** Additional figures, TGA, powder X-ray diffraction patterns, IR spectra, sorption isotherms, and

CIF files. This material is available free of charge via the Internet at http://pubs.acs.org.

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